

Supplementary Information for
Large-Scale Computational Screening of Binary Intermetallics for Membrane-based Hydrogen Separation

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1. Overview of method

The following figure illustrates the algorithm of the method developed in this work to screen hundreds of intermetallic structures without compromising on the accuracy of our predictions.

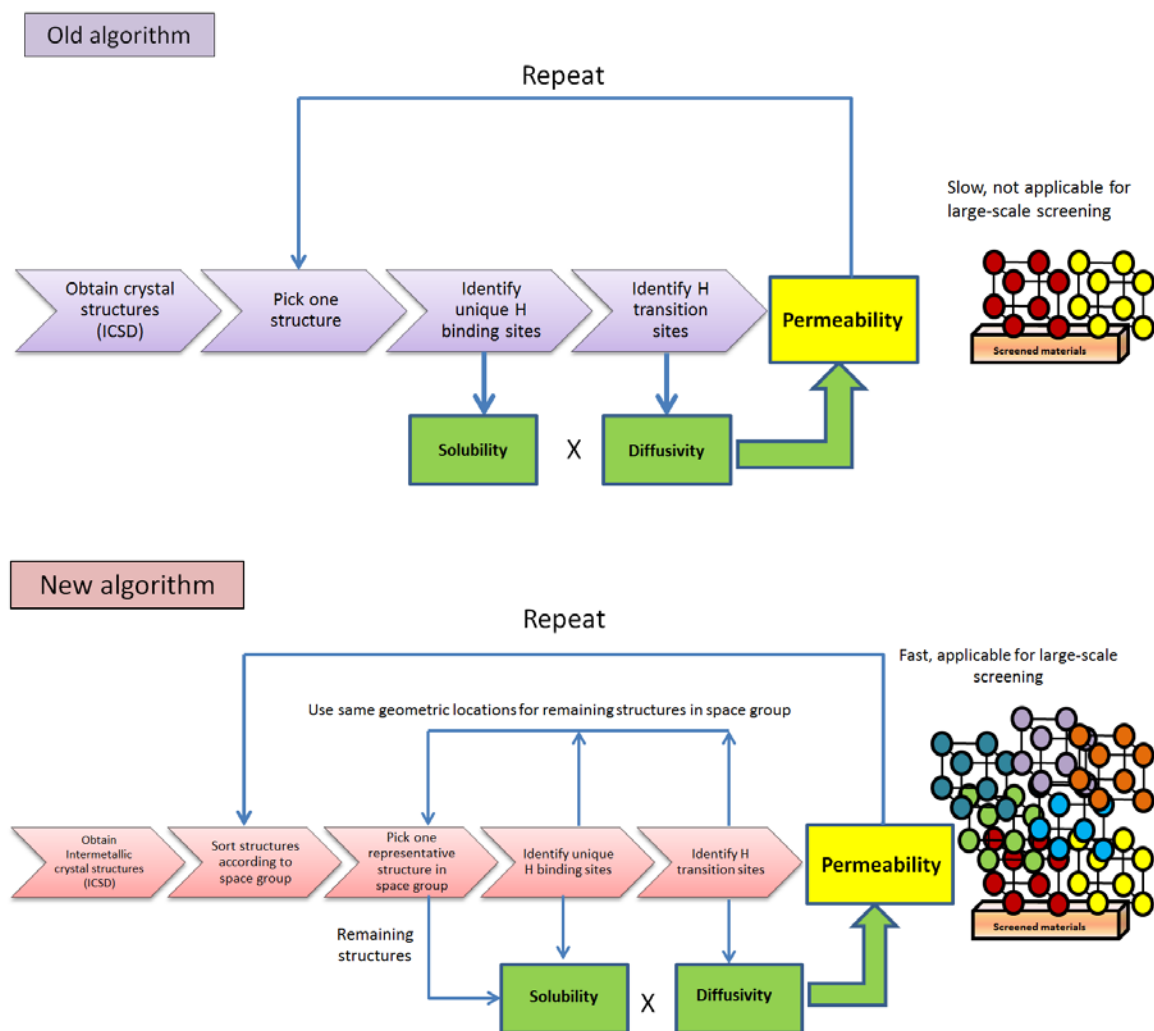


Figure S1: A schematic comparison of the old method¹ and the new algorithm developed to rapidly screen favorable materials for H separation from 981 binary intermetallics. This method exploits the geometries of the different structures to

reduce the total number of DFT calculations needed to evaluate properties enabling the screening of orders of magnitude more materials than possible previously. This method is easily automated using python libraries such as PYMATGEN and is more efficient than previous methods of computational prediction.

2. Screening Algorithm for non-Pd based intermetallics

In the main text, the screening process for identifying non-Pd based membranes from ICSD². In this section, the selection process for identifying suitable H membrane materials from thousands of intermetallics listed in ICSD is explained in detail. A search in the ICSD for binary intermetallics that are a combination of alkali and alkaline earth metals, all d-block metals and Al, Ga, In, Sn, Tl, Pb and Bi from the p-block lists 1625 unique materials. Structures for which no specific melting temperatures or pressures were provided in ICSD were also included in this study. When multiple entries existed for the same material that listed identical crystal structures and lattice constants, the most recent entry was used for further analysis.

From these entries, the materials unsuitable for membrane applications due to various factors were removed. The different criteria used to remove materials and the number of such entries removed from the list of candidates is listed in Table S1.

Table S1: List of the various conditions used to sort the candidate materials. The number of materials satisfying each condition is listed on the right.

Condition	No.
All ICSD entries	1625
Materials with multiple phases	127
Materials with $T_{\text{melt}} < 1500$ K	104
Materials stable only at high pressures (> 10 atm)	15
Materials stable only above 500 K	89
Purely computational structures	35
Remaining candidates	1255

There are 127 materials that show multiple phases i.e. undergo a phase transition at some temperature from 400-1200 K. For example, the intermetallic NiPt undergoes a change in structure from P4/mmm to Fm-3m at 1000 K. For these materials the individual phases were treated as separate materials at the conditions specified. For example, at temperatures < 1000 K, the calculations to predict solubility of NiPt were performed on its P4/mmm structure, while the Fm-3m structure was used at temperatures > 1000 K. The H solubility of each of the phases of the 127 materials was calculated individually for the relevant temperature ranges. These materials were

included in the analysis of potential candidate materials and a total of 1382 materials (1255+127) were screening using the method described in the main text.

Materials that are stable only at high temperatures (> 500 K) or high pressures (> 10 atm) were removed from the list. Many intermetallics had melting points < 1500 K. For membrane purposes, thermal stability at high temperatures is essential since gas streams containing a mixture of hydrogen and other gases can be produced at 1093 K or higher temperatures³. Hence materials with melting points (T_{melt}) lower than 1500 K were removed from the list of potential candidates.

We did not include intermetallic phases whose existence was reported purely computationally. It is possible that other intermetallic structures exist that have not yet been discovered or reported in the ICSD. Their identification however, is beyond the scope of this work.

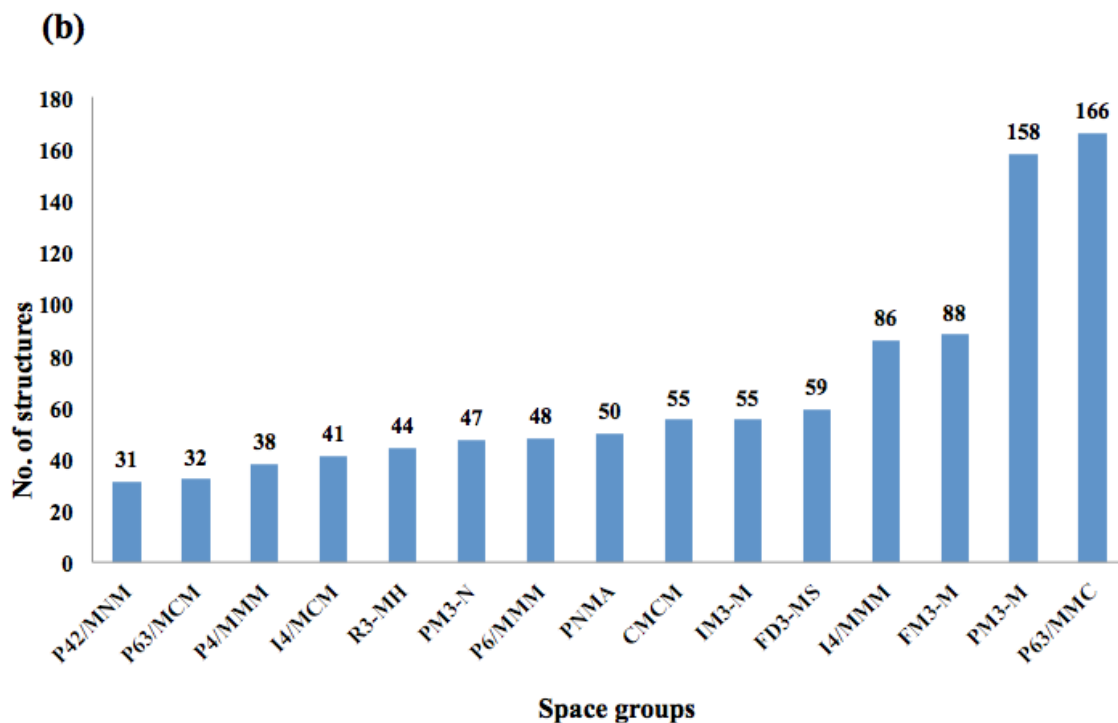
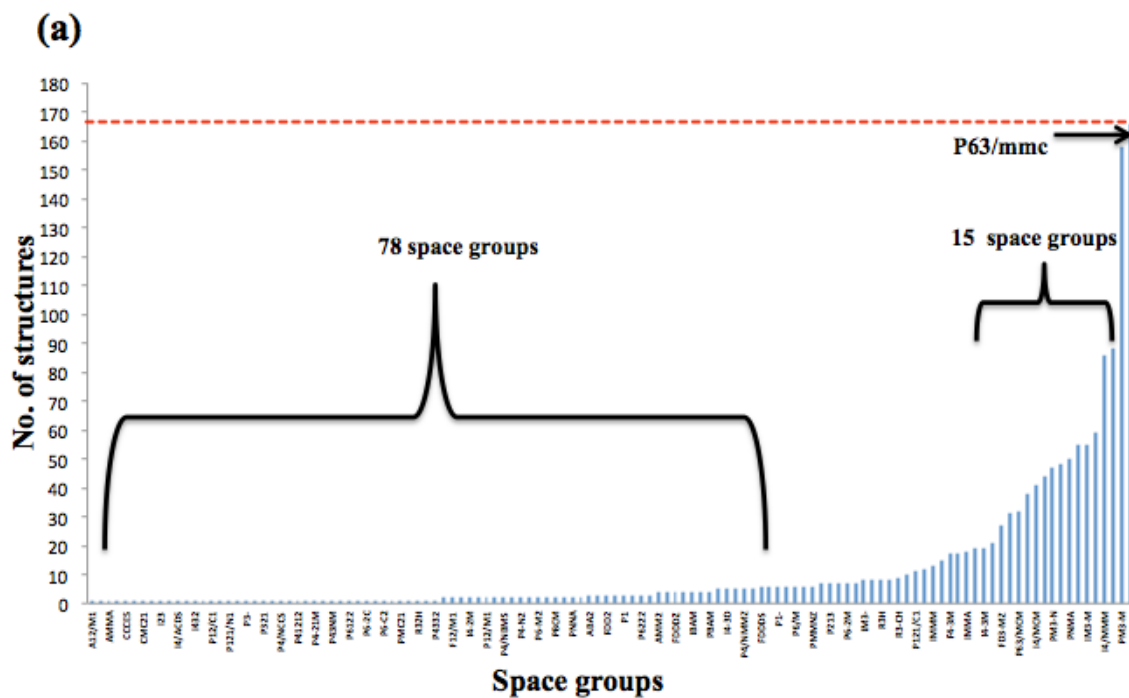


Figure S2: (a) 1255 intermetallic materials sorted into 123 space groups. The horizontal axis represents the space groups and the vertical axis is the number of materials in each space group. 78 space groups having 5 materials or less are

highlighted. P63/mmc is the space group with the maximum number of materials (166), shown by the red dashed line. For reasons of clarity, not all space groups are labeled on the horizontal axis. (b) The top 15 most populated space groups considered in this work.

From Fig. S2, we can observe that there are some space groups such as P63/mmc that have a large number of candidate structures (166) and there are 78 space groups that have 5 or less candidate structures. To make the screening process efficient, the space groups were ranked based on the number of candidate structures of each type in the space group and the top 15 were selected for the next steps in the screening process. A substantial proportion of structures (64%) belong to these 15 groups. These 15 space groups include 981 materials that meet the other criteria listed in Table S1.

To rapidly identify favorable candidates from the 981 materials in 15 space groups using solubility as a screening parameter and a minimum number of DFT calculations, a representative structure was chosen randomly from each space group. The top 15 space groups and the representative material chosen for each space group are shown in Table S2. The total number of structures that belong to the 15 space groups is 995. This includes 14 materials that show two phases where both phases belong to the 15 space groups. This results in 981 unique materials (removing the double counted materials). In the case of Cu_3Sn , one of the structures belongs to the top 15 space groups (P63/mmc) however the other low temperature structure belongs to F4-3m space group which is not one of the 15 space groups. Similarly HfV_2 has two other low temperature phases (Fd-3mz and Imma) apart from Fd-3ms, which are not included in these 15 space groups. NiTi shows three other structures apart from its Pm-3m structure i.e. P-, P121/m1 and P1121/m. For these three materials the 6 structures that do not belong to the 15 most populated space groups were also considered and the calculations were performed separately for the relevant temperatures. In total, in this work, calculations were performed on 1001 (995+6) structures.

During the solubility calculation using the method described in the main text unique H binding sites in each material are identified. We simplified our analysis by assuming that the geometry of these unique binding sites is common between crystal structures within the same space group. This assumption is consistent with data from previous calculations for different Fm-3m (fcc) structures that resulted in consistent geometric locations of the H-binding sites^{4,5}. Although the locations of binding sites are the same in similar materials, the binding energy of H in each material depends on the material's constituent elements.

Table S2: The top 15 most common space groups along with the number of structures of each space group type are shown. The prototype column lists the composition of the intermetallic selected as the representative structure of the space group.

Rank	Space Group	No. of structures	Representative Structure
1	P63/mmc	166	Fe ₂ Nb ₁
2	Pm-3m	158	Co ₁ Ti ₁
3	Fm-3m	88	Ir ₁ Sn ₂
4	I4/mcm	86	Ba ₅ Pb ₃
5	Fd-3ms	57	Be ₂ Cu ₁
6	Cmcm	55	Bi ₂ Ca ₁
7	Im-3m	55	Nb ₁ Hf ₁
8	Pnma	50	Au ₁ Y ₂
9	P6/mmm	47	Fe ₅ Y ₁
10	Pm-3n	47	Au ₁ Zn ₃
11	R-3mh	44	Ir ₃ Y ₁
12	I4/mmm	41	Au ₂ Hf ₁
13	P4/mmm	38	Cd ₃ Zr ₁
14	P63/mcm	32	Hf ₅ Ir ₃
15	P42/mnm	31	Cr ₂ Ru ₁

To ensure that DFT calculations give accurate lattice constants for the different materials considered, the lattice constants of the structures obtained from the ICSD[1] were compared with those obtained from DFT. The crystal structures obtained from the ICSD were used as input in geometric relaxation calculations performed using DFT. 169 of the candidate structures obtained from ICSD had partial occupancies or incomplete structures and could not be directly used as an input for DFT calculations. For these structures, the positions of the atoms were obtained from prototype structures and the lattice constants listed in the ICSD were used to construct a unit cell. For 158 structures, the resulting lattice parameters predicted by DFT calculations were in good agreement with the lattice constants listed in the ICSD. For the remaining 11 structures a difference in cell volume > 5% was found. These 11 structures are listed in Table S2 marked with an asterisk. In the remaining 832 structures whose structure files from the ICSD were directly used in DFT calculations, 25 structures had lattice constants that result in cell volumes that differ by greater than 5% from the values listed in ICSD. These 25 structures are listed in Table S3

(without the asterisk). For these structures, the DFT optimized structures were used for further calculations. The lattice constants calculated using DFT for the remaining 807 structures obtained directly from ICSD are in good agreement with the ICSD values. We used the DFT optimized structures for all materials as a starting point for the next steps in the screening process.

Material	ICSD values	ICSD volume	Calculated lattice parameters	DFT volume	DFT vol./ ICSD vol.
InSn ₄	a=3.21,c= 3.00, $\alpha=\beta=90^\circ,\gamma=120^\circ$	26.81	a=3.31,c= 3.56, $\alpha=\beta=90^\circ,\gamma=120^\circ$	47.21	1.45
Pt ₂ Re ₃ *	a=2.77,c= 4.43, $\alpha=\beta=90^\circ,\gamma=120^\circ$	29.32	a=2.96,c= 4.73, $\alpha=\beta=90^\circ,\gamma=120^\circ$	46.16	1.41
NiRu	a=2.61, c=4.20, $\alpha=\beta=90^\circ,\gamma=120^\circ$	24.70	a=2.74, c=4.6, $\alpha=\beta=90^\circ,\gamma=120^\circ$	38.81	1.40
Ga ₂ Sr ₁ *	a=4.35,c= 4.74, $\alpha=\beta=90^\circ,\gamma=120^\circ$	77.55	a=4.45,c= 5.15, $\alpha=\beta=90^\circ,\gamma=120^\circ$	120.71	1.32
Mg ₄₇ Pb ₃ *	a=3.21,c=5.42, $\alpha=\beta=90^\circ,\gamma=120^\circ$	46.86	a=3.26,c=5.58, $\alpha=\beta=90^\circ,\gamma=120^\circ$	82.37	1.27
In ₉ Pb	a=3.27,c=5.03, $\alpha=\beta=\gamma=90^\circ$	53.82	a=3.57,c=5.25, $\alpha=\beta=\gamma=90^\circ$	66.95	1.24
CrFe*	a=2.9 , $\alpha=\beta=\gamma=90^\circ$	24.39	a=3.1 , $\alpha=\beta=\gamma=90^\circ$	29.79	1.22
Ti ₇ Zr ₃ *	a=3.03,c= 4.83, $\alpha=\beta=90^\circ,\gamma=120^\circ$	38.40	a=3.05,c= 5.03, $\alpha=\beta=90^\circ,\gamma=120^\circ$	46.80	1.22
IrPt*	a=3.88 , $\alpha=\beta=\gamma=90^\circ$	58.28	a=4.12 , $\alpha=\beta=\gamma=90^\circ$	90.52	1.20
CoV ₃ *	a=4.68 , $\alpha=\beta=\gamma=90^\circ$	102.24	a=4.98 , $\alpha=\beta=\gamma=90^\circ$	123.51	1.20
TaTi	a=3.29 , $\alpha=\beta=\gamma=90^\circ$	35.48	a=3.49 , $\alpha=\beta=\gamma=90^\circ$	59.32	1.19
Co ₈ Sn ₄₄ *	a=12.14 , $\alpha=\beta=\gamma=90^\circ$	1787.11	a=12.84 , $\alpha=\beta=\gamma=90^\circ$	2708.87	1.18
Cr ₁₃ Ga ₁₃	a=12.63,c=7.79, $\alpha=\beta=90^\circ,\gamma=120^\circ$	1074.61	a=12.75,c=7.89, $\alpha=\beta=90^\circ,\gamma=120^\circ$	1269.24	1.18
Pt ₃ V ₂ *	a=3.86 , $\alpha=\beta=\gamma=90^\circ$	57.56	a=4.05 , $\alpha=\beta=\gamma=90^\circ$	91.13	1.16
Cu ₄ Ti ₃	a=3.13,c=19.97, $\alpha=\beta=\gamma=90^\circ$	195.09	a=3.35,c=20.11, $\alpha=\beta=\gamma=90^\circ$	225.68	1.16
Ta ₆ Zn ₇	a=5.04,c=27.53, $\alpha=\beta=90^\circ,\gamma=120^\circ$	604.37	a=5.24,c=27.73, $\alpha=\beta=90^\circ,\gamma=120^\circ$	697.87	1.15
Be ₁₇ Zr ₂	a=7.54,c=11.02, $\alpha=\beta=90^\circ,\gamma=120^\circ$	542.04	a=7.68,c=11.15, $\alpha=\beta=90^\circ,\gamma=120^\circ$	625.89	1.15
Mo ₁₇ Pt ₃ *	a=4.89 , $\alpha=\beta=\gamma=90^\circ$	124.18	a=5.20 , $\alpha=\beta=\gamma=90^\circ$	191.10	1.13
AuBe ₁₂	a=7.243,c=4.25, $\alpha=\beta=\gamma=90^\circ$	223.06	a=7.45,c=4.55, $\alpha=\beta=\gamma=90^\circ$	252.54	1.13
CdLi ₃ *	a=4.26 , $\alpha=\beta=\gamma=90^\circ$	77.25	a=4.41 , $\alpha=\beta=\gamma=90^\circ$	122.02	1.11
NbTa	a=3.3, $\alpha=\beta=\gamma=90^\circ$	35.94	a=3.42, $\alpha=\beta=\gamma=90^\circ$	54.87	1.11
Na ₇ Ga ₁₃	a=15.63,b=14.98,c=21.7, $\alpha=\beta=\gamma=90^\circ$	5073.67	a=15.9,b=15.65,c=22.5, $\alpha=\beta=\gamma=90^\circ$	5598.79	1.10
Co ₇ Cr ₈	a=b=8.81,c=4.56, $\alpha=\beta=\gamma=90^\circ$	353.93	a=b=8.91,c=4.86, $\alpha=\beta=\gamma=90^\circ$	385.83	1.09
Cs ₄ Sn ₂₃	a=b=c=12.10, $\alpha=\beta=\gamma=90^\circ$	1769.80	a=b=c=12.43, $\alpha=\beta=\gamma=90^\circ$	1920.50	1.09

Material	ICSD values	ICSD volume	Calculated lattice parameters	DFT volume	DFT vol./ ICSD vol.
Ca ₇ Sn ₆	a=7.87,b=23.81,c=8.465, $\alpha=\beta=\gamma=90^\circ$	1586.14	a=7.98,b=24.02,c=8.9, $\alpha=\beta=\gamma=90^\circ$	1705.81	1.08
Fe ₂₃ Y ₆	a=b=c=12.08, $\alpha=\beta=\gamma=90^\circ$	1763.67	a=12.35, $\alpha=\beta=\gamma=90^\circ$	1883.65	1.07
Sn ₁₀ Y ₁₁	a=11.53,c=16.91, $\alpha=\beta=\gamma=90^\circ$	2248.03	a=11.83,c=17.02, $\alpha=\beta=\gamma=90^\circ$	2381.93	1.06
Ba ₅ Pb ₃	a=9.04,c=16.81, $\alpha=\beta=\gamma=90^\circ$	1374.23	a=9.24,c=17.09, $\alpha=\beta=\gamma=90^\circ$	1459.10	1.06
Nb ₃ Os ₂	a=b=9.86,a=5.06, $\alpha=\beta=\gamma=90^\circ$	492.02	a=b=9.88,c=5.32, $\alpha=\beta=\gamma=90^\circ$	517.00	1.05
Ba ₁₁ Bi ₁₀	a=b=13.21,c=19.37, $\alpha=\beta=\gamma=90^\circ$	3389.51	a=b=13.41,c=19.58, $\alpha=\beta=\gamma=90^\circ$	3565.77	1.05
Pb ₄ Sr ₅	a=8.48,b=17.27,c=9.01, $\alpha=\beta=\gamma=90^\circ$	1319.51	a=8.58,b=17.47,c=9.07, $\alpha=\beta=\gamma=90^\circ$	1385.49	1.05

Table S3: Comparison of the experimental and the DFT calculated structural parameters of intermetallics for structures where there was > 5% deviation in cell volume from the reported ICSD structures. All distances are in Angstroms and volume in Å³. The structures marked with an asterisk are the structures for which the structures were generated manually and not using the cif files from ICSD.

Table S4: List of intermetallic compounds and the energy of the most favorable H binding site. The solubility for these materials cannot be reliably calculated using Sieverts' law. These materials are not favorable for use as membranes.

Material	E _b (eV)	Material	E _b (eV)
Ti ₇ Zr ₃	-4.78	Ga ₂ Y ₃	-0.96
Ga ₄ Ti ₆	-4.36	Ga ₃₉ Na ₂₂	-0.92
IrTi ₃	-4.03	Be ₅ Hf	-0.92
BiK ₃	-3.56	Li ₃ Bi	-0.91
Ga ₃ Zr	-3.45	Ca ₅ Ga ₃	-0.91
Ga ₃ Nb ₅	-3.38	CaGa ₂	-0.91
In ₂ Li ₃	-2.8	GaLi	-0.91
GaHf ₂	-2.77	BaMg ₂	-0.9
Bi ₁₀ Ca ₁₁	-2.7	CsRb	-0.89
MgIn	-2.6	Mg ₂ Sn	-0.87
Ga ₃ Hf ₅	-2.56	Be ₂ Zr	-0.87
LiZn	-2.49	Mg ₃₈ Sr ₉	-0.87
Ga ₅ Ni	-2.4	Ga ₁₄ Li ₃	-0.86
Bi ₁₀ Sr ₁₁	-2.3	CdLi	-0.86
Ga ₄ Nb ₅	-1.93	Ca ₁₁ Ga ₇	-0.85
Ga ₃ Ti ₅	-1.86	AuNa ₂	-0.84
CaNi ₂	-1.78	Ga ₃ Hf ₁	-0.81
NaGa ₄	-1.75	Ga ₃ Sc ₅	-0.78
Mg ₂₃ Sr ₆	-1.7	Ca ₁₃ Cd ₇₆	-0.76
LiTl ₁	-1.65	Li ₃ Tl	-0.76
HfTi	-1.62	CaMg ₂	-0.76
Ga ₄ Ti ₅	-1.6	Ga ₃ Nb ₂	-0.73
In ₃ Li ₁₃	-1.6	Ga ₂ Li ₃	-0.71
Ga ₃ Nb	-1.57	CaSr	-0.71
InNa	-1.57	Ga ₁₀ Hf ₁₁	-0.67
CdLi ₃	-1.56	TiZr	-0.67
BaGa ₄	-1.56	K ₈ Sn ₄₆	-0.65
Li ₃ Pd	-1.54	KPb ₂	-0.65
Ga ₃ K ₂	-1.54	Li ₅ Tl ₂	-0.61
Li ₂ Sr ₃	-1.5	Na ₁₃ Pb ₅	-0.59
Sr ₆ Li ₂₃	-1.46	HfSc	-0.57
CaLi ₂	-1.35	Mg ₂ Pb	-0.56
MgPd ₃	-1.34	GaZr ₂	-0.56
Ga ₄ Sr	-1.34	LiPt ₂	-0.54
LiPt ₇	-1.2	Ca ₃₁ Sn ₂₀	-0.52
Be ₁₇ Hf ₂	-1.2	BaLi ₄	-0.52
LiMg	-1.2	Mg ₃ In	-0.47
GaTi ₂	-1.2	BiSr ₂	-0.47
MgSc	-1.07	Ba ₆ Mg ₂₃	-0.45
Ga ₅ Mg ₂	-1.04	Ba ₁₀ Ga	-0.42
Be ₂ Hf	-1.04	Ag ₃ Ca ₅	-0.38
MgRh	-1.04	Cs ₆ K ₇	-0.37
Ga ₇ Li ₂	-1.03	NaTl	-0.35

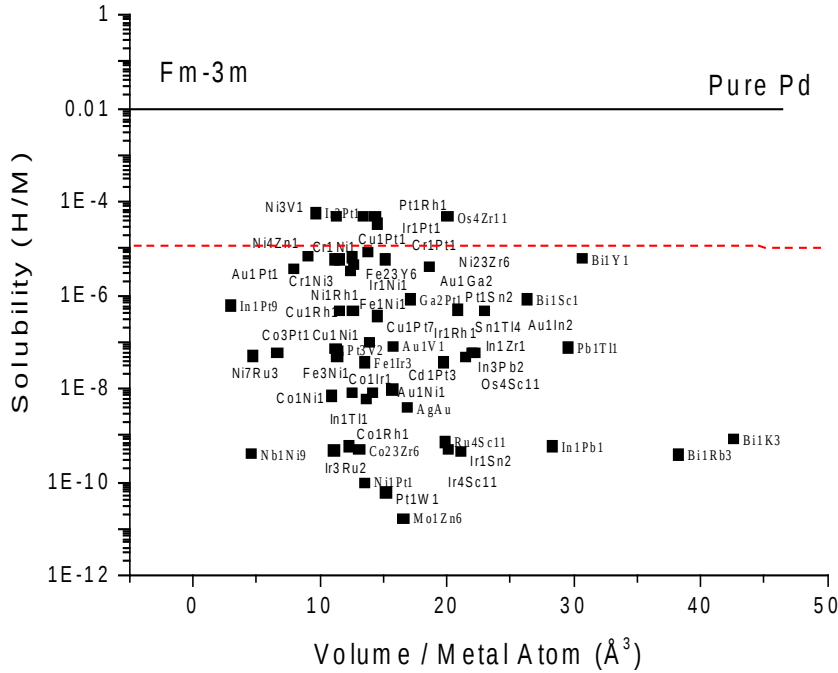
HfOs	-1.03	BiCa ₂	-0.33
GaY	-1.03	Au ₂ Na	-0.33
LiPd ₇	-0.98	In ₄ K	-0.3

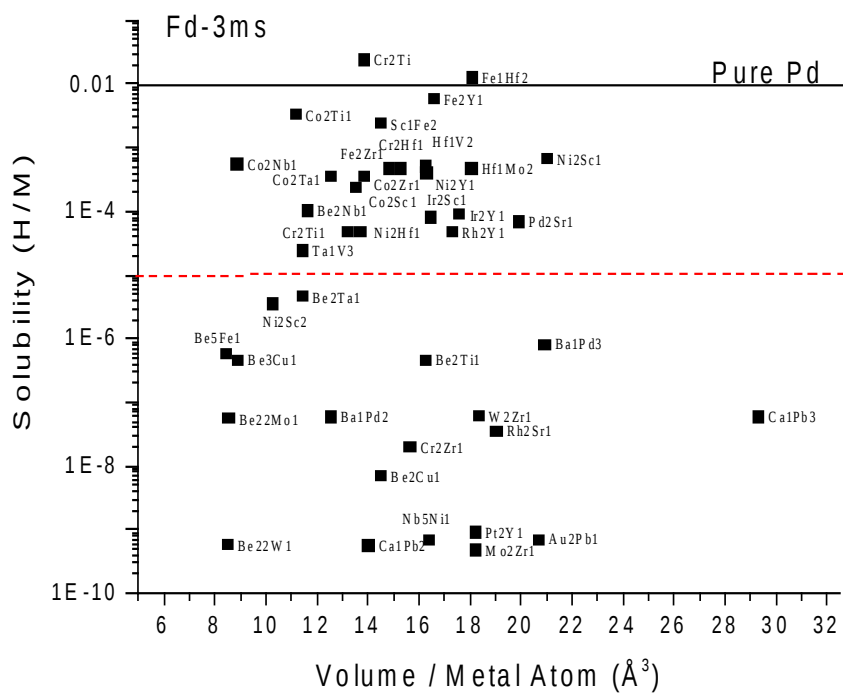
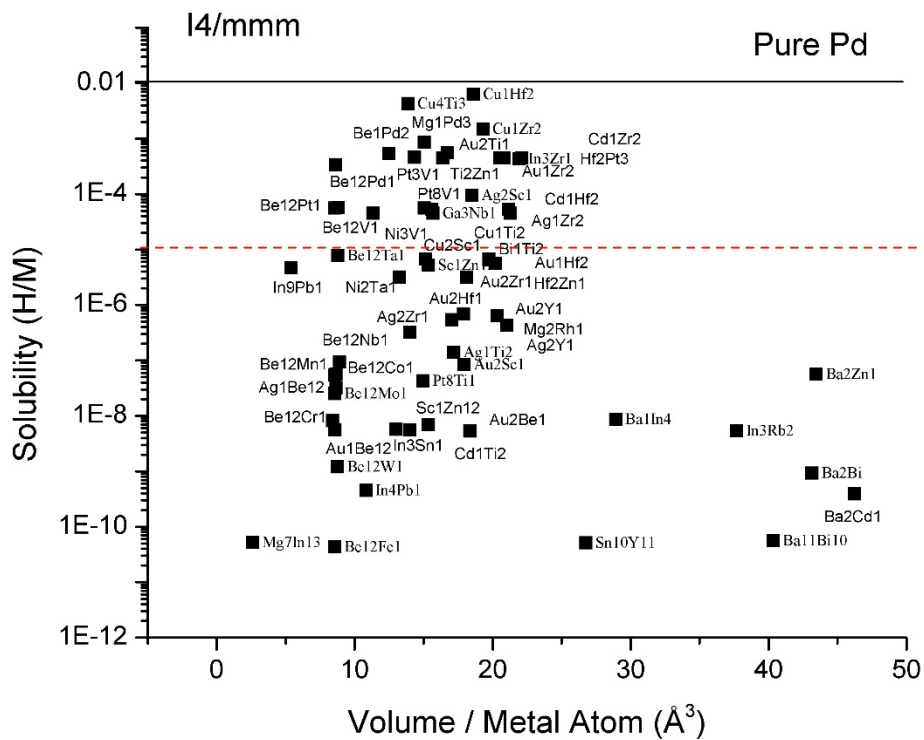
3. Solubility as a screening parameter

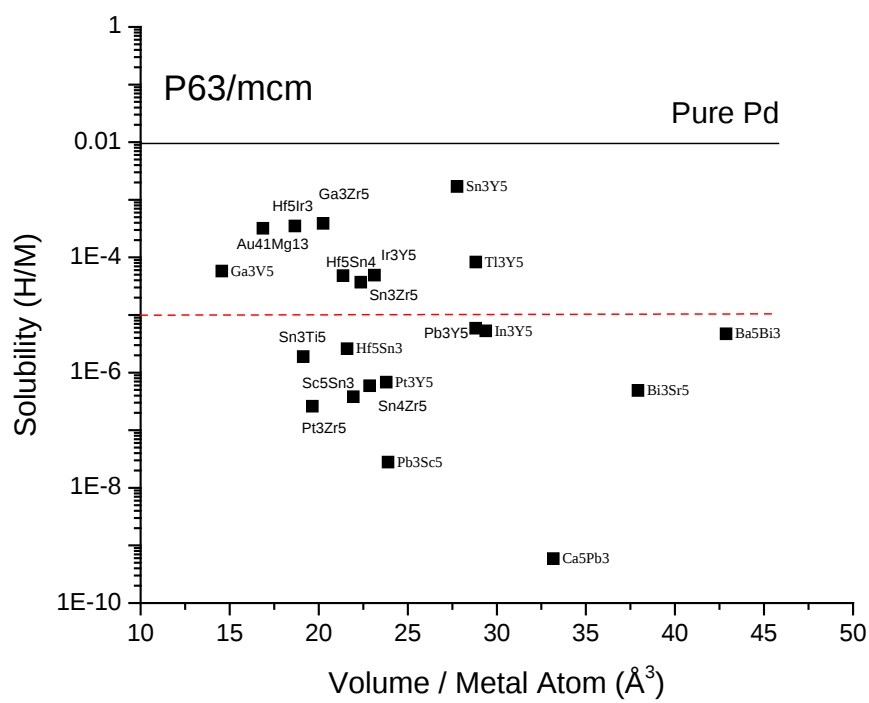
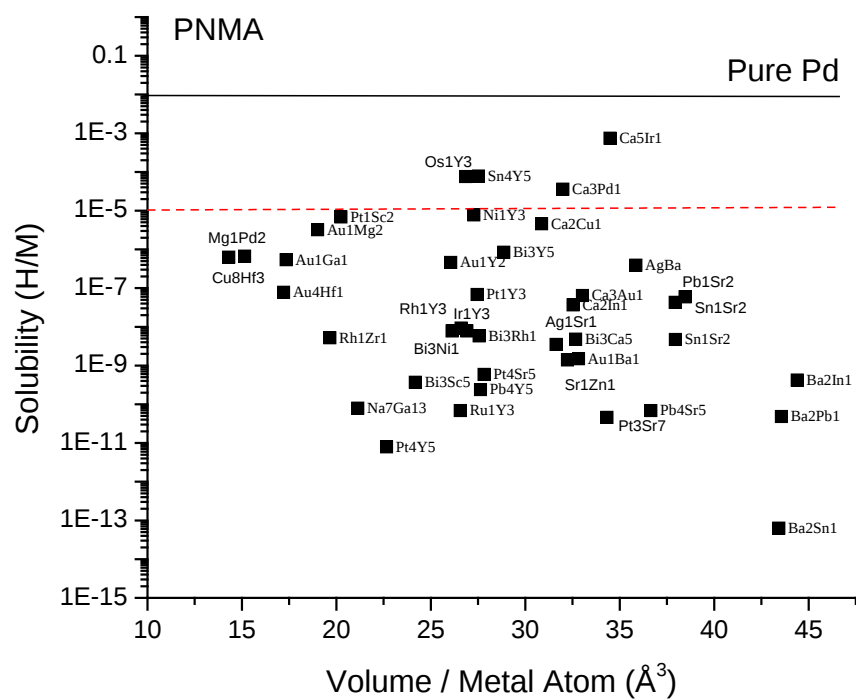
As described in the main text, solubility was used as a screening parameter to identify suitable intermetallic membrane materials for H separation. The solubility based screening was showing for Cmcm space group as an example in the main text. The remaining 14 space groups are shown below.

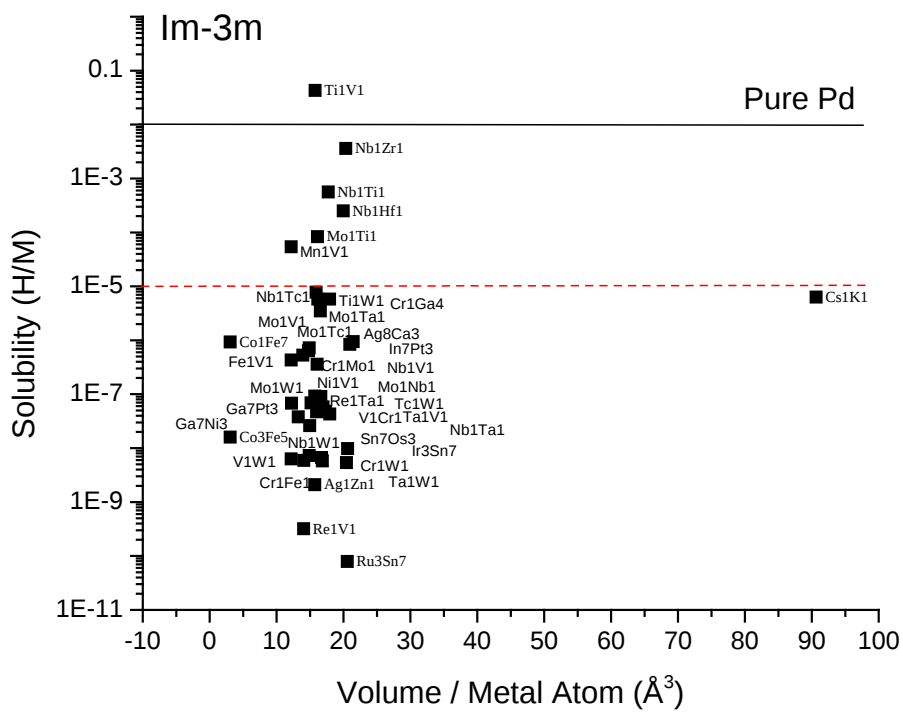
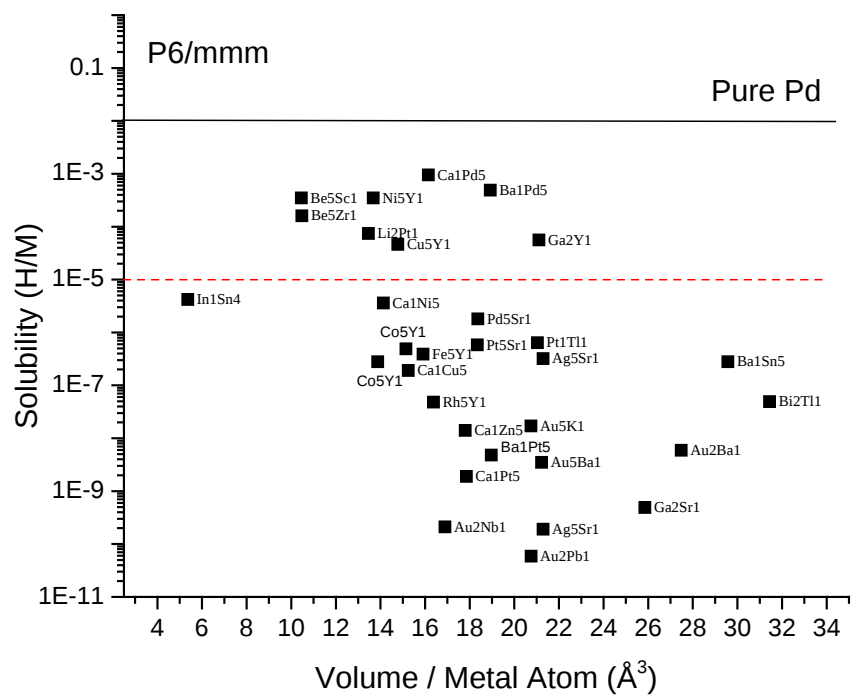
The diffusion activation energy of the shortlisted materials from each of the space groups using the solubility as a screening parameter is shown in table S5.

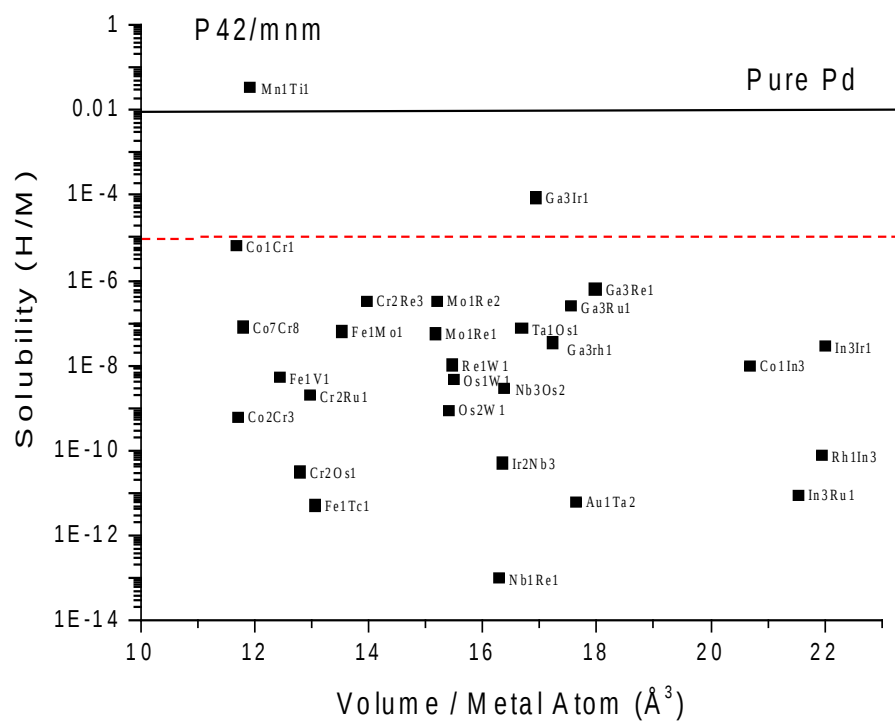
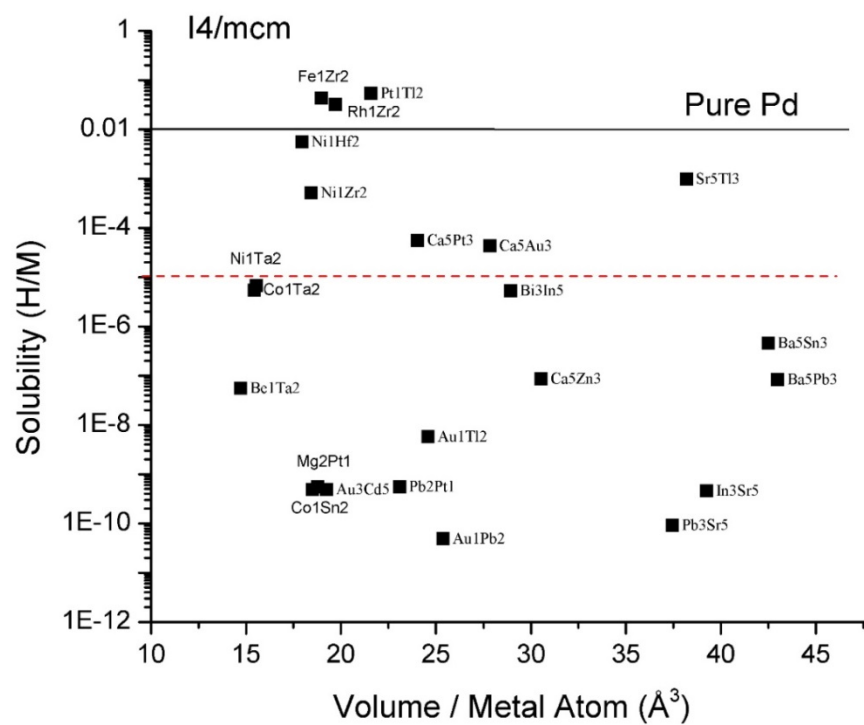
Figure S3: Solubility (H/M) of hydrogen at 600 K and 1 atm H₂ in intermetallics belonging to a specific space group. Each data point is labeled by the respective intermetallic. The solubility (H/M) for pure Pd computed with the same methods is shown as a horizontal solid line for comparison. The volume/metal atom for pure Pd is 15.9 Å³. The black solid line and the red dashed line show the favorable solubility range used to screen candidates (10⁻¹-10⁻⁵ H/M). Intermetallics belonging to 14 space groups are shown here, with the identity of the space group indicated on each figure.

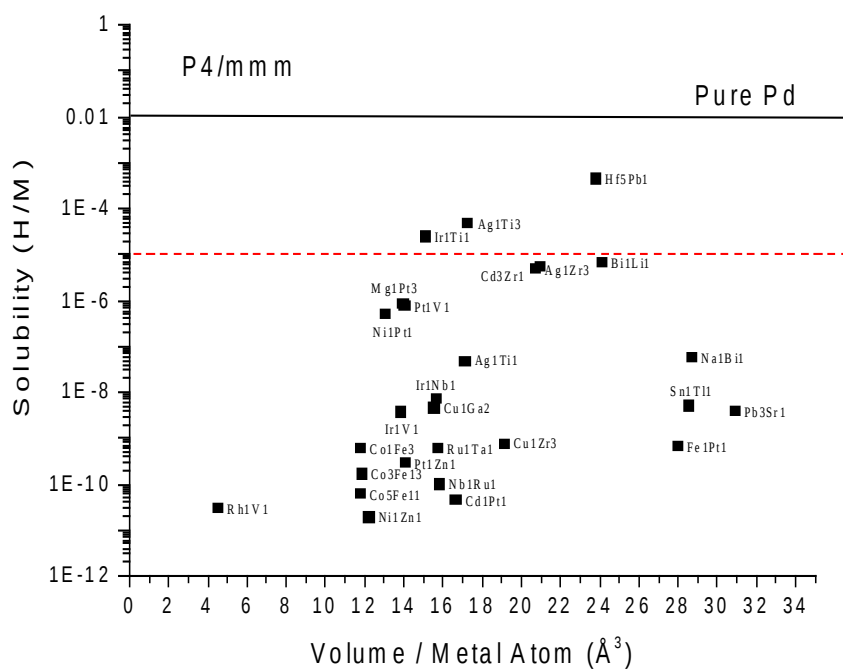
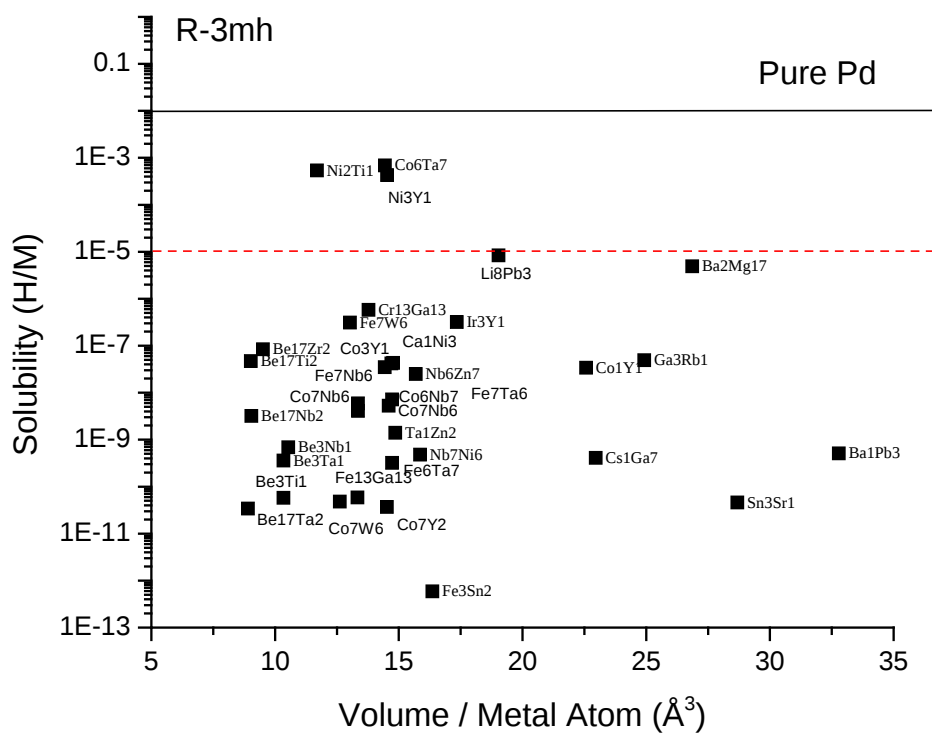


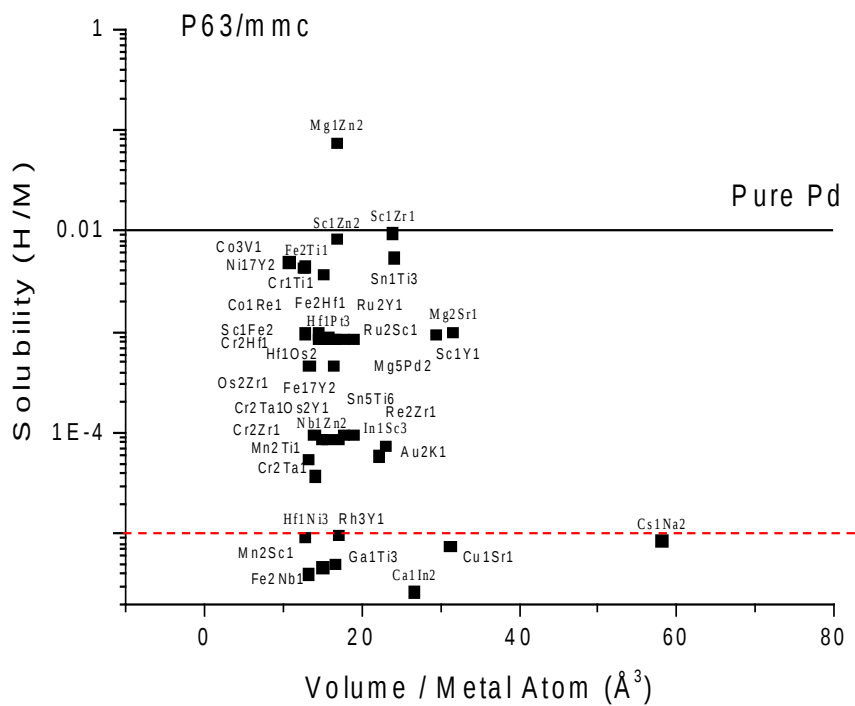
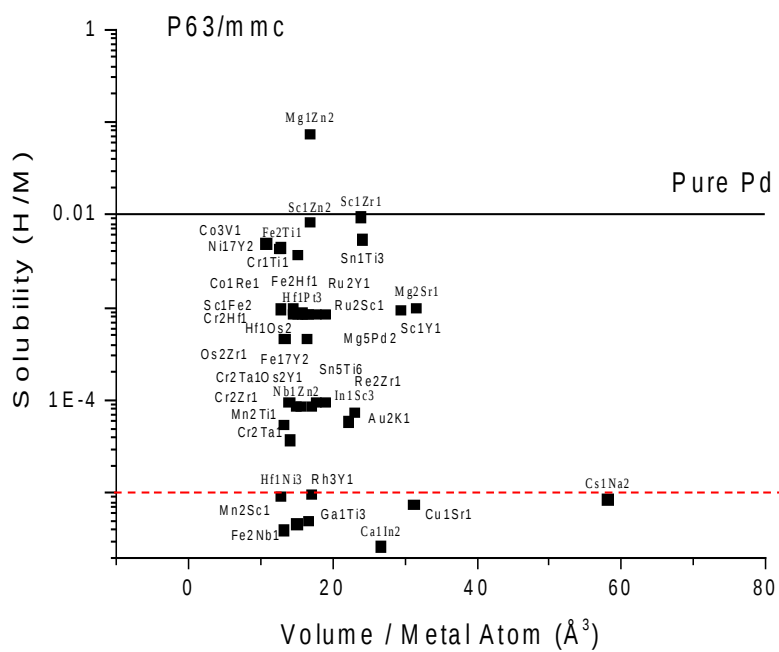












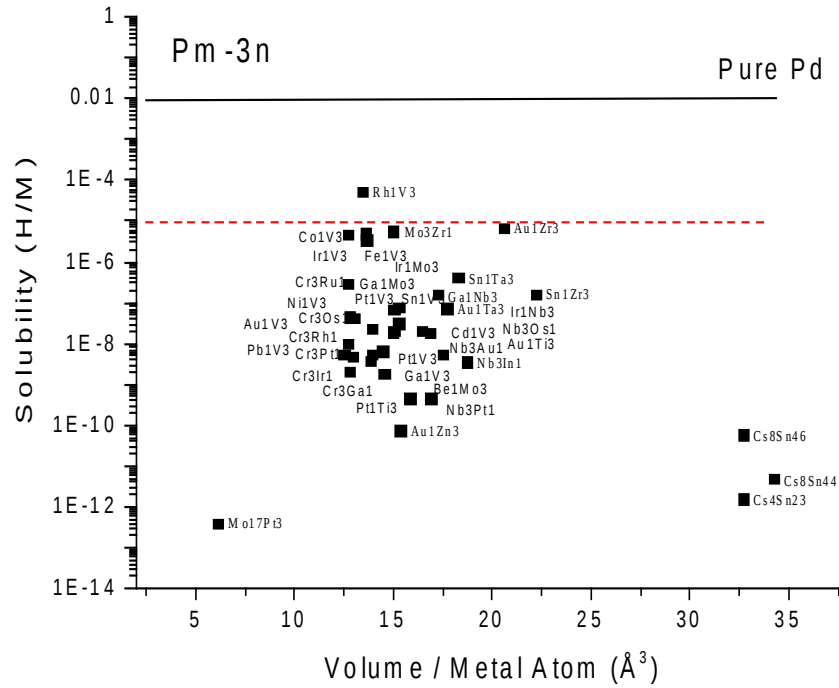
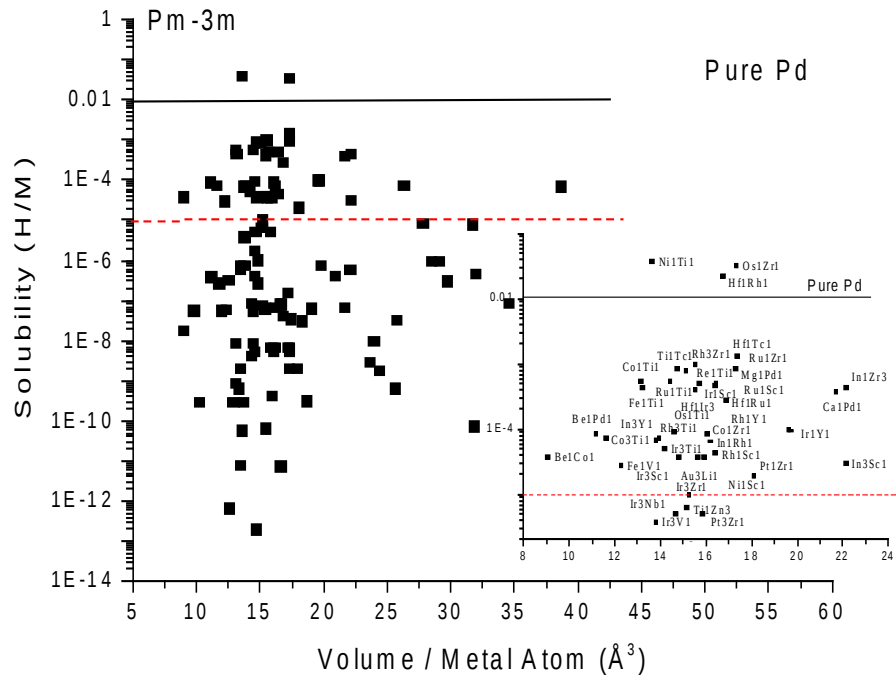


Table S5: Ensemble average binding energies (Eb) and diffusion activation energies (Ea) of shortlisted intermetallics at 600 K and 1 atm H₂. All values are in eV.

Intermetallic	Eb	Ea	Intermetallic	Eb	Ea
Sc1Fe2	-0.07	0.19	Ru2Y1	-0.02	0.39
Rh1V3	0.13	0.21	Ti2Zn1	0.01	0.39
Ni17Y2	-0.12	0.22	Hf5Pb1	0.02	0.39
Pt8V1	0.12	0.22	Pt1Rh1	0.13	0.39
Fe2Y1	-0.22	0.27	Ag1Zr2	0.13	0.39
Mn1V1	0.12	0.23	Cr1Ti1	-0.12	0.4
Mg1Zn2	-0.22	0.27	Be1Pd2	0.01	0.4
Hf1Ta1	-0.13	0.28	Cr1Ti1	-0.13	0.4
Sc1Fe2	-0.07	0.24	Ni3V1	0.13	0.4
Ni2Ti1	0.01	0.25	Ni3V1	0.13	0.4
Fe2Hf1	-0.02	0.26	Rh1Sc1	0.13	0.4
Fe1Ti1	0.02	0.26	Ru2Sc1	-0.02	0.41
Mn1Ti1	-0.19	0.24	Hf5Sn4	0.13	0.41
Hf1V2	0.01	0.27	Ti1Zn16	0.13	0.41
Cs1Au1	0.11	0.27	Pt1Zr1	0.17	0.41
Ru1Zr1	-0.02	0.28	Fe1Zr2	-0.21	0.42
Hf1Mo2	0.01	0.28	Cr2Ti	-0.18	0.42
Sn4Y5	0.10	0.28	Cr2Hf1	-0.02	0.42
Mn2Ti1	0.12	0.28	Nb1Zn2	0.10	0.42
Pb2Y1	0.13	0.28	Tl3Y5	0.10	0.42
Ir1Ti1	0.16	0.28	Ta1V3	0.16	0.42
Pt1Ti2	-0.23	0.29	Rh5Zr3	0.01	0.43
Fe1Hf2	-0.15	0.29	Co1Ti1	0.01	0.43
Nb1Ti1	0.01	0.29	Ta1Ti1	0.01	0.43
Ni3Y1	0.02	0.29	Rh1Y1	0.09	0.43
Hf1Ni1	0.05	0.29	Mo1Ti1	0.10	0.43
Ni1Hf2	-0.14	0.3	Be12Pt1	0.12	0.43
Nb1Zr1	-0.13	0.3	Ir3Y5	0.13	0.43
Pt8V1	0.12	0.3	Rh2Y1	0.13	0.43
Ni2Hf1	0.13	0.3	Sc1Zr1	-0.14	0.44
Co6Ta7	-0.01	0.31	Sn3Y5	-0.05	0.44
Hf1Os2	0.01	0.31	Ni2Sc1	0.00	0.44
Ir1Zr1	0.02	0.32	Li2Pt1	0.10	0.44
Be5Sc1	0.03	0.32	Be12V1	0.12	0.44
Co2Sc1	0.05	0.32	Co2Nb1	0.01	0.45
In3Y1	0.11	0.32	Co2Zr1	0.03	0.45
Cu1Ti2	0.12	0.32	Cr2Ta1	0.09	0.45
Cu5Y1	0.13	0.32	Au2K1	0.12	0.45
Ir1Pt1	0.15	0.32	Ir3Ti1	0.12	0.45
Mg1Pd1	0.01	0.33	Ti1Tc1	-0.02	0.46
Hf5Ir3	0.03	0.33	Ir1Y1	0.09	0.46
Os1Y3	0.10	0.33	In1Rh1	0.11	0.46
Pd2Sr1	0.11	0.33	Ga2Y1	0.12	0.46
Cu1Pt1	0.13	0.33	Os4Zr11	0.13	0.46
Ba1Pd1	0.16	0.33	Ca5Au3	0.13	0.46
Co3V1	-0.12	0.34	Sr5Ti3	-0.02	0.47

Intermetallic	Eb	Ea	Intermetallic	Eb	Ea
Ga3Zr5	0.02	0.34	Ca5Ir1	-0.01	0.47
Ni1Ti1	-0.22	0.30	In1Sc3	0.11	0.47
Os2Zr1	-0.02	0.35	Rh3Ti1	0.11	0.47
Be12Pd1	0.03	0.35	Cr2Ti1	0.13	0.47
Ir2Y1	0.10	0.35	Ca3Pd1	0.14	0.47
Ga3Ir1	0.10	0.35	Cr2Hf1	-0.02	0.48
Ir2Pt1	0.12	0.35	Re1Ti1	-0.01	0.48
Ti1V1	-0.18	0.30	Ir1Sc1	0.01	0.48
Cu1Zr2	-0.05	0.36	Ni2Y1	0.02	0.48
Hf1Pt3	-0.02	0.36	Hf1Ru1	0.04	0.48
Ni1Zr2	0.01	0.36	Sc1Y1	-0.02	0.49
Cd1Zr2	0.01	0.36	Fe17Y2	0.01	0.49
Fe2Zr1	0.01	0.36	In1Zr3	0.02	0.49
Ni1Zr1	0.02	0.36	Ca1Pd1	0.03	0.49
Hf1Mo1	0.03	0.36	Ca5Pt3	0.12	0.5
Ag1Ti3	0.13	0.36	Ca5Pt3	0.12	0.5
In3Sc1	0.15	0.36	Sn3Zr5	0.14	0.5
Cu4Ti3	-0.14	0.367	Ir2Sc1	0.10	0.51
Fe2Ti1	-0.13	0.37	Ag2Sc1	0.09	0.52
Pt2Ta1	0.00	0.37	Hf1Tc1	-0.04	0.53
Ni5Y1	0.03	0.37	Ni1Sc1	0.14	0.53
Nb1Hf1	0.04	0.37	Cr2Zr1	0.10	0.54
Be5Zr1	0.07	0.37	Be1Co1	0.14	0.54
Os2Y1	0.09	0.37	Hf1Ir3	0.02	0.55
Be1Pd1	0.10	0.37	Tl1Y1	0.11	0.55
Sc1Zn2	-0.13	0.38	Co1Zr1	0.10	0.56
Cu1Hf2	-0.12	0.38	Ga3V5	0.12	0.56
Co2Ti1	-0.12	0.38	Mg5Pd2	-0.02	0.57
Mg2Sr1	-0.02	0.38	Sn5Ti6	0.09	0.61
Ru1Sc1	0.01	0.38	Be2Nb1	0.09	0.63
Mo1Zr1	0.02	0.38	Ca1Pd5	-0.02	0.64
Co2Ta1	0.03	0.38	Ru1Ti1	0.01	0.64
Re2Zr1	0.10	0.38	Co3Ti1	0.11	0.64
Cd1Hf2	0.12	0.38	Au41Mg13	0.03	0.65
Ir3Sc1	0.14	0.38	Os1Ti1	0.10	0.66
Rh1Zr2	-0.20	0.39	Au3Li1	0.14	0.67
Co1Re1	-0.02	0.39	Os1Zr1	-0.20	0.78
Rh3Zr1	-0.02	0.39	Sn1Ti3	-0.13	0.86

4. Diffusivity and Solubility normalized to pure Pd of the selected 161 materials.

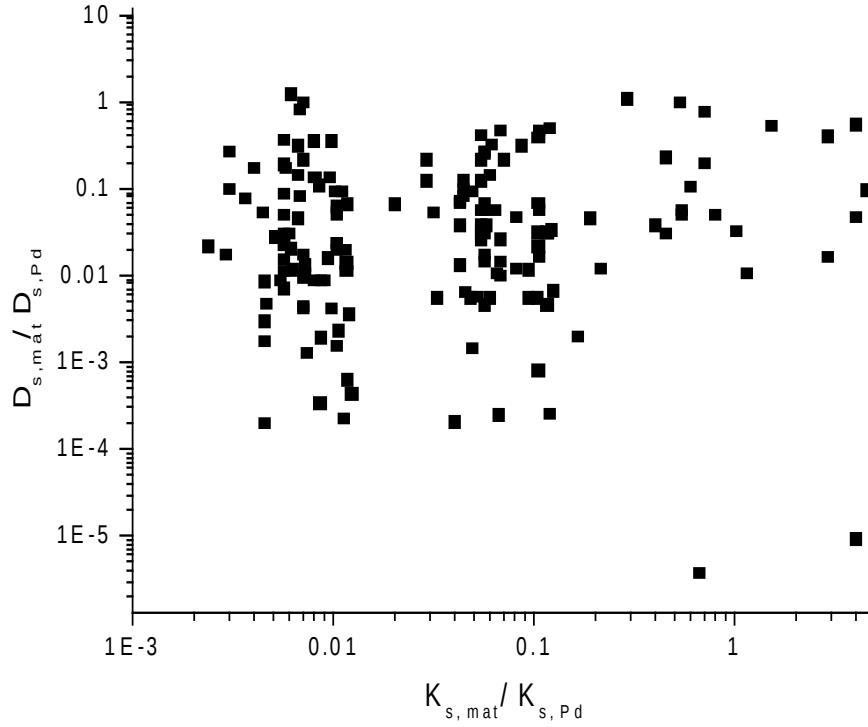


Figure S4: Normalized diffusivity as a function of normalized solubility for 161 intermetallics. The reference system is pure Pd. All properties were calculated at 600 K and 1 atm H_2 .

5. Mn-Ti-H equilibrium phase diagram

Using the procedure developed by Alapati and Sholl^{6,7} to evaluate the stability of phases for combinations of up to four elements and hydrogen, we predicted the phase diagram of Mn-Ti-H at a H_2 pressure of 1 atm. Figure S4 shows the predicted Mn-Ti-H phase diagram.

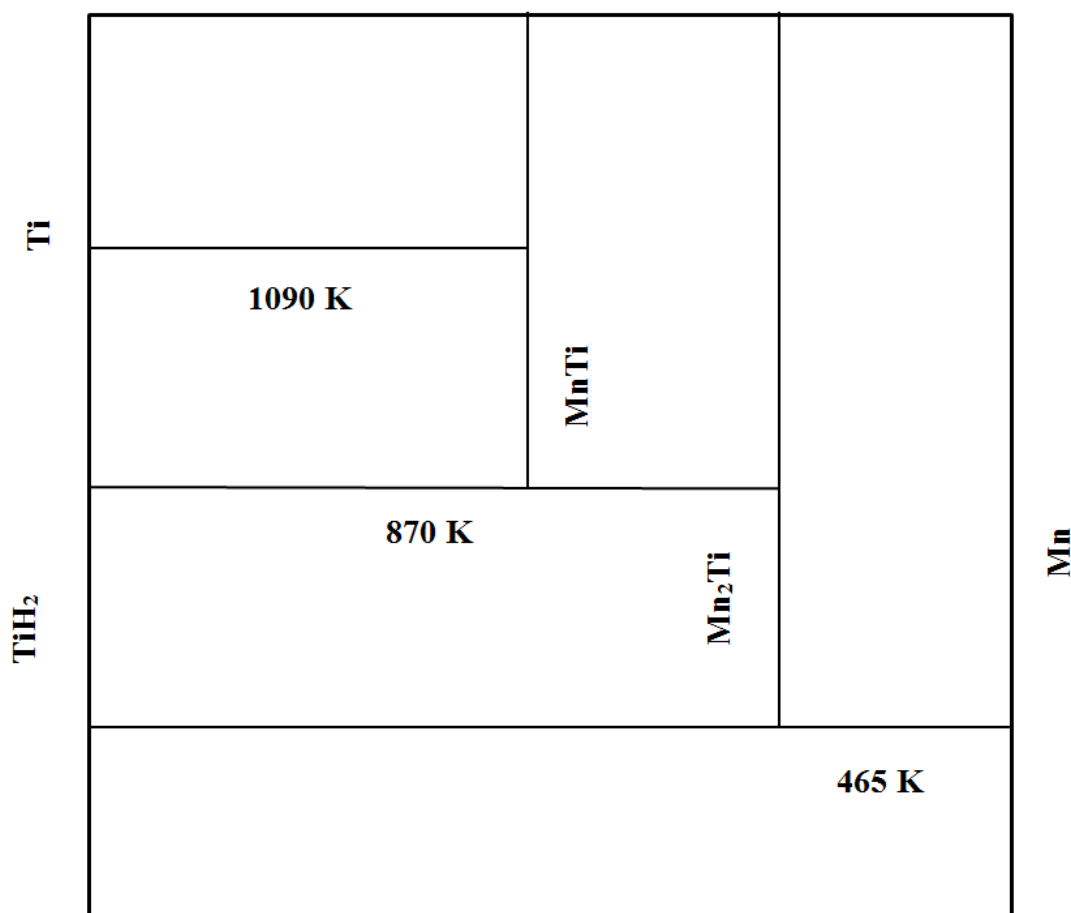


Figure S5: Mn-Ti-H calculated equilibrium phase diagram at 1 atm H₂. The decomposition temperatures of various phases are shown using horizontal lines. The vertical lines indicate the compounds formed. No entropic (vibration effects) were considered in these calculations.

According to our calculations, MnTi is formed at temperatures greater than 870 K. At lower temperatures, MnTi separates to form Mn₂Ti (870-465 K) and further dissociates into the more stable Mn and TiH₂ phases. These calculations were performed at 0 K and do not include vibrational effects. Inclusion of vibrational effects typically lowers the decomposition temperatures but does not qualitatively affect the predicted phase diagram⁸⁻¹⁰.

References

- 1 Kamakoti, P. *et al.* Prediction of hydrogen flux through sulfur-tolerant binary alloy membranes. *Science* **307**, 569-573 (2005).
- 2 Bergerhoff, G. & Brown, I. D. *Crystallographic Databases*. (International Union of Crystallography, 1987).
- 3 Ockwig, N. W. & Nenoff, T. M. Membranes for Hydrogen Separation. *Chemical Reviews* **107**, 4078-4110 (2007).
- 4 Kang, S. G., Coulter, K. E., Gade, S. K., Way, J. D. & Sholl, D. S. Identifying Metal Alloys with High Hydrogen Permeability Using High Throughput Theory and Experimental Testing. *The Journal of Physical Chemistry Letters* **2**, 3040-3044 (2011).
- 5 Chandrasekhar, N. & Sholl, D. S. Quantitative computational screening of Pd-based intermetallic membranes for hydrogen separation. *J. Membr. Sci.* **453**, 516-524 (2014).
- 6 Alapati, S. V., Johnson, J. K. & Sholl, D. S. Large-scale screening of metal hydride mixtures for high-capacity hydrogen storage from first-principles calculations. *Journal of Physical Chemistry C* **112**, 5258-5262 (2008).
- 7 Ozolins, V., Majzoub, E. H. & Wolverton, C. First-Principles Prediction of Thermodynamically Reversible Hydrogen Storage Reactions in the Li-Mg-Ca-B-H System. *Journal of the American Chemical Society* **131**, 230-237 (2008).
- 8 Nicholson, K. M. & Sholl, D. S. Computationally efficient determination of hydrogen isotope effects on the thermodynamic stability of metal hydrides. *Phys. Rev. B* **86**, 134113 (2012).
- 9 Nicholson, K. M. & Sholl, D. S. First-Principles Screening of Complex Transition Metal Hydrides for High Temperature Applications. *Inorganic Chemistry* **53**, 11833-11848 (2014).
- 10 Nicholson, K. M. & Sholl, D. S. First-Principles Prediction of New Complex Transition Metal Hydrides for High Temperature Applications. *Inorganic Chemistry* **53**, 11849-11860 (2014).